



# **STREEM™ – InGaN Edition**

**STRain Engineering in Electronic Materials**

**User Guide**

**December 2014**



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## 1. Introduction

Nowadays, III-nitride electronic and optical devices (LEDs, laser diodes, transistors, etc.) are actively developed in order to achieve higher performance at lower cost that would expand their areas of applications. Aside from the device design, much effort is focused on tuning and refining the conditions and recipes used to grow epitaxial structures with desired properties. Optimization of the growth conditions and choice of the best epitaxy procedure providing high quality materials with targeted composition and doping is rather nontrivial, especially in the case of III-nitride semiconductors. Varying such parameters as the reactor pressure, carrier gas composition, precursor flow rates, and growth temperature, as well as the use of special procedures like growth interruptions, pre-depositions, and temperature ramping is known to impact strongly the performance of nominally the same device structures. Depending on particular recipe and structure design, modifications of the growth conditions may induce changes in compositional and structural properties of the grown crystals, affecting distributions of strain and defect density in the grown heterostructures. In turn, this may lead to dramatic changes in optical and electrical device characteristics. Detailed understanding and quantification of correlations between the growth recipe and heterostructure properties is a key factor providing successful optimization of epitaxial technology.

STREEM-InGaN is a specialized software tool for modeling the characteristics of (0001) III-Nitride device heterostructures grown by MOCVD from conventional metal-organic precursors (TMIn, TMGa/TEGa, TMAI) and ammonia, diluted in  $H_2/N_2$  carrier gases. STREEM-InGaN focuses on an InGaN-based active region which implies a sequence of quantum wells and barriers as well as other stages in-between. Layers grown prior to and after the active region can be added into the simulations as well.

The software is aimed at understanding and control of the structure properties by adjusting the process recipe. In particular, the following issues can be addressed:

- influence of the process parameters on indium incorporation into the quantum wells;
- predictions of the actual composition profile in the active region of the heterostructure, including delayed indium incorporation into the QWs and indium tails in the cap layers or barriers. Due to indium surface segregation, the actual composition profile normally deviates from the nominal one built up from the steady-state solutions obtained for every individual epilayer at the respective growth conditions;
- consistent computations of indium incorporation and elastic energy allows the users to follow and adjust the strain distribution in the active region by both modifying the operating parameters for the



particular layers and adding strain-relief layers underneath the active region. The actual composition and strain profiles determine the distribution of the polarization charges in the structure that can be accounted for in subsequent modeling of device operation with the SiLENSe software [<http://www.str-soft.com/products/SiLENSe/>];

- the onset of stress relaxation via formation of V-shaped dislocation half-loops and subsequent evolution of the strain, threading dislocation density, and indium composition profile can be studied with the STREEM-InGaN software, depending on the particular parameters in the recipe.

As a result of the modeling, the user can analyze such characteristics as the growth rate and composition profile across the heterostructure and the distributions of strain and dislocation density. By adjusting the recipe parameters, including temperature, pressure, flow rates of the precursors and carrier gases, as well as the sequence and durations of the particular stages of the process, the user can follow the respective changes in the above characteristics and establish correlations between the recipe and properties of the heterostructure.

The user is welcome to contact STR by e-mail [STREEM-Support@str-soft.com](mailto:STREEM-Support@str-soft.com) with any problems or questions about STREEM operation.



## 2. Installation and licensing

### 2.1 Installation

To install the STREEM-InGaN software, run the installation executable file and follow the installer instructions. To uninstall the program, run the **unins000.exe** file in the program home directory or use the respective item in the *Programs* menu. Note that the software needs write permission for its home directory.

After the installation, the STREEM-InGaN package includes the following items:

- **STREEM-InGaN.exe** executable file and dll files required for software operation;
- **Doc** folder that contains the User Guide;
- **Examples** folder that contains a number of example cases;
- Files required for licensing the software

The software is supplied with the license protection system based on a dongle key which is a small USB device. During installation, the user is prompted to install the dongle key driver.

### 2.2 Single node license

The user can install the program on several PCs (for example, on the desktop and notebook) and run the program on the PC where the dongle key is plugged in. The registration procedure is as follows:

- Plug in the dongle key module;
- Start the license manager (License.exe) located in the STREEM-InGaN home directory. One can also do it by Windows Start button;
- Choose the *Key config* tab and press the *Make Key* button. The request file (**\*.req**) containing the information about the dongle will be generated;
- Send the generated request file to STR by e-mail [STREEM-support@str-soft.com](mailto:STREEM-support@str-soft.com);
- When the reply from STR is received, plug in the dongle, start the license manager, press *Accept Key* button, and browse for the license file (**\*.lic**) received from STR;
- Now one can run STREEM-InGaN.

Note that the maximum number of simultaneous runs of the software is limited as specified in the particular license agreement (quotation, purchase order, etc.).



### 2.3 Corporate/department network license

Corporate/department license allows the user to run the software on several computers within the local area network (LAN). The software may be installed on any computer of the network, but the maximum number of simultaneously running copies of the software is limited as specified in the particular license agreement (quotation, purchase order, etc.).

Hereafter, by the 'server' we will understand the computer where the USB dongle key is plugged in. The user should install the dongle key driver on this PC. A special tool **server.exe** called below 'license server' to be running on the server computer. After start, the respective icon appears in the system tray. The pop-up menu allows one to restore the window and inspect the event log or the license server. The *Change Port* button allows the user choose the port number for communication between the software running on other PCs and the license server.

To obtain/update the license, one needs to run the **License.exe** file on the server PC. Further procedure is similar to that described in Section 2.2.

All the PCs running the STREEM-InGaN software we will call 'workstations'. Running the software on the server computer is also allowed. Installation of the software on the workstations is described in Section 2.1. The user does not need to install the dongle key driver on workstations. Instead, one should run the **License.exe** file, choose *Server config* tab, and specify the IP address of the server PC and the port number. The *Store* button saves this information in the **Str\_Config\_SL.cfg** file in the software home directory. This file is used by the software to communicate with the license server. The *Remove* button clears the file. The *Test* button allows one to test the communication with the license server. The following conditions to be fulfilled for running the software on a certain workstation: (i) the dongle key to be plugged into the server PC, (ii) license server to be running on the server PC, and (iii) the workstation is able to communicate with the license server by the IP address and port number specified in the **Str\_Config\_SL.cfg** file in the software home directory. Unplugging the dongle key from the server PC or stopping the license server will result in abnormal termination of all the running software copies.

### 3. Theoretical background

#### 3.1 Governing equations to describe composition profile in device heterostructures

The underlying surface kinetic model considers MOCVD of InGaN/GaN heterostructure from typical MO compounds TMIn, TMGa, and TEGa as the group-III precursors and NH<sub>3</sub> as the N precursor. The growth occurs in the atmosphere of carrier gas that is a mixture of N<sub>2</sub> and possibly H<sub>2</sub>. The structure is grown via diffusion of the precursors through the concentration boundary layer that is formed above the growth surface and determined by the gas species transport in particular MOCVD reactors. The incoming gas species decompose at the growth surface to produced adsorbed atoms of In, Ga, N, and H, which partly desorb back to the gas in the form of the initial gas species and monatomic In and Ga and partly incorporate into the crystal to produce ternary solid solution InGaN.

The model utilizes an approach that was suggested and further developed in Refs. [1-3]. The approach is based on the unsteady balance of the adsorbed species that is described by the kinetic equations as follows.

$$\frac{d\theta_{In}}{dt} = \sum_g J_g \nu_{g,In} - J_{InN} \quad (1)$$

$$\frac{d\theta_{Ga}}{dt} = \sum_g J_g \nu_{g,Ga} - J_{GaN} \quad (2)$$

$$\frac{d\theta_N}{dt} = \sum_g J_g \nu_{g,N} - J_{InN} - J_{GaN} \quad (3)$$

$$\frac{d\theta_H}{dt} = \sum_g J_g \nu_{g,H} \quad (4)$$

$$\frac{X}{1-X} = \frac{J_{InN}}{J_{GaN}} \quad (5)$$

Here,  $\theta_a$  are the surface molar densities of the adsorbed atoms ( $a = \text{In, Ga, N, and H}$ ),  $J_g$  are the species molar fluxes between the gas and adsorbed layer,  $\nu_{g,a}$  are the stoichiometric coefficients,  $J_{InN}$  and  $J_{GaN}$  are the species molar fluxes between the adsorbed layer and the crystal, and  $X$  is the InN molar fraction in InGaN.



The gas-adsorbate fluxes are found as

$$J_g = \frac{P_g^0 - K_g \prod_a (\theta_a)^{\nu_{g,a}}}{R_g} \quad (6)$$

Here,  $P_g^0$  are the species partial pressures at the conventional external border of the concentration boundary layer (i.e. at the interface between the boundary layer and the flow core),  $K_g$  are the reduced desorption rate constants, and  $R_g$  are the transport resistances between the flow concentration core and the growth surface.

Species partial pressures can be found as  $P_g^0 = PG_g / G$ , where  $P$  is the total pressure,  $G_g$  are the species input molar fluxes, and  $G = \sum_g G_g$  is the total input molar flux. The species transport resistances are given as

$$R_g = R_g^d + R_g^k \quad (7)$$

where  $R_g^d = RT\delta / D_g$  are the diffusion transport resistances,  $R_g^k = 1 / \alpha_g \beta_g$  are the kinetic (Knudsen) transport resistances;  $D_g$  are the diffusivities,  $R$  is the gas constant,  $T$  is the temperature,  $\delta$  is the concentration boundary layer thickness,  $\alpha_g$  are the sticking coefficients, and  $\beta_g$  are the Hertz-Knudsen factors.

The adsorbate-solid fluxes are found as

$$J_{InN} = K_{InN}^+ \theta_{In} \theta_N - K_{InN}^- \gamma_{InN} \chi_V X, \quad J_{GaN} = K_{GaN}^+ \theta_{Ga} \theta_N - K_{GaN}^- \gamma_{GaN} (1 - X) \quad (8)$$

where  $K_{InN}^+$  and  $K_{GaN}^+$  are the rate constant of InN and GaN incorporation into the crystal,  $K_{InN}^-$  and  $K_{GaN}^-$  are the rate constants of InN and GaN decomposition,  $\gamma_{InN}$  and  $\gamma_{GaN}$  are the activities of InN and GaN constituents of InGaN solid solution. The latter quantities determine deviation of InGaN from the ideal solution and are related to two factors: mixing energy of the constituents and stress in the structure. The InGaN growth rate is determined as

$$V_g = J_{InN} W_{InN} + J_{GaN} W_{GaN} \quad (9)$$

where  $W_{InN}$  and  $W_{GaN}$  are the molar volumes of the corresponding solid constituents.



Kinetic constants are related to each other due to the detailed balance principle by the conditions of equilibrium, corresponding to the three overall surface reactions



Analysis of numerous literature data suggests that formation of indium profile in the InGaN-based heterostructures is largely limited by the desorption of atomic In from the growth surface and thus determined by the In desorption rate constant, which is approximated as a function of temperature. Sticking probability of low-reactive  $\text{N}_2$  is approximated as a function of temperature and In molar fraction  $\alpha_{\text{N}_2}(T, X)$  while those of the other species, except maybe for the In-containing ones, are assumed unity. Sticking probabilities of the In-containing species depend on the capacity of In adsorbed layer. Two models of the In adsorbed layer are incorporated into the software, one limits its capacity to one monolayer, while the other does not limit it. Thus, the two models correspond to the lower and upper limits of the In adsorbed layer capacity. In the former case, the sticking probabilities of the In-containing species are not unity due to site blocking with adsorbed indium and can be found as

$$\alpha_g = \chi_v = 1 - \theta_{\text{In}} / \theta_s \quad (10)$$

where  $\chi_v$  is the surface coverage with vacancies and  $\theta_s$  is the surface molar density of the In adsorption sites.

Given the process recipe, i.e. the growth temperature and the species flow rates as functions of time,  $T(t)$  and  $G_g(t)$ , the developed kinetic model represents a closed stiff set of five differential-algebraic equations (1)-(5) in five unknown functions of time,  $\theta_a(t)$  and  $X(t)$ . The software solves it numerically using the Radau method with automatic adaptation of time steps, which is just aimed at the solution of stiff problems. With the derived solution and supplementary relationships (6)-(9), the gas-adsorbate fluxes, the adsorbate-crystal fluxes, and the growth rate are also found as functions of time,  $J_g(t)$ ,  $J_{\text{InN}}(t)$ ,  $J_{\text{GaN}}(t)$ , and  $V_g(t)$ . Then, the interface displacement relative to its initial location is given as

$$z(t) = \int_0^t V_g(t) dt \quad (11)$$

Eventually, the desirable In content profile in an InGaN/GaN structure  $X(z)$  is implicitly determined by

two functions of time, they are  $X(t)$  and  $z(t)$ .

### 3.2 Contribution of stress and mixing energy to the InGaN growth chemistry

Stress and mixing energy affect indium incorporation into the structure via deviation of realistic InGaN solid solution from the ideal solution. The effect is associated with a change of equilibrium constants, corresponding to the overall surface reactions, due to changes of chemical potentials of the solid solution constituents, i.e. InN and GaN. Chemical potentials of the gas species little differ from those of ideal gases

$$\mu_g = \mu_g^0(T) + RT \ln(P_g / P_s) \quad (12)$$

where  $\mu_g^0(T)$  are the temperature-dependent chemical potentials under the standard pressure  $P_s$ . Simultaneously, chemical potentials of the solid solution constituents essentially differ from those in the ideal solution due to the considerable mixing energy and elastic stress. Following the approach developed in Ref. [4], these potentials are found as

$$\begin{aligned} \mu_{InN} &= \mu_{InN}^0(T) + RT \ln(X) + W(1 - X^2) + \mu_{InN}^s \\ \mu_{GaN} &= \mu_{GaN}^0(T) + RT \ln(1 - X) + WX^2 + \mu_{GaN}^s \end{aligned} \quad (13)$$

where  $W$  is the mixing energy and  $\mu_{InN}^s$  and  $\mu_{GaN}^s$  are the contributions of the stress.

The two latter quantities account for two effects: elastic strain arising due to the difference between the crystal lattice constants at the growth surface and at the last underlying layer (i) and plastic strain induced by V-shaped dislocation half-loops (HLs), which are expected to be generated during the growth of layers in the active region (ii). As it is shown in Ref. [5], formation of such dislocations represents the major mechanism of stress relaxation in (0001) InGaN/GaN heterostructures, consistent with the main experimental findings and providing reasonable agreement of calculated and experimental data on the critical InGaN layer thickness.

Quantities  $\mu_{InN}^s$  and  $\mu_{GaN}^s$  represent explicit functions of the lattice and elastic constants of the constituents (i) and parameters of the plastic strain, namely, surface density of HLs  $\rho$ , angle of their opening  $\alpha$ , and coordinate of their apex  $z_0$  (ii). The two latter parameters are assumed to be the same for all the HLs. The opening angle  $\alpha$  is considered as an input parameter of the problem, while quantities  $\rho$  and  $z_0$  are found from the energy balance similar to that used in Ref. [5] to derive the

critical thickness for the generation of dislocations and in Ref. [6] to derive the conditions for the inclination of threading dislocations in AlGaIn. Here, the “energetic criterion” is given as

$$\Delta E(z, z_0, \rho) = E(z, z_0) - W(z, z_0, \rho) = 0 \quad (14)$$

where  $E(z, z_0)$  is the self-energy of the V-shaped dislocation HLs and  $W(z, z_0, \rho)$  is the energy release of the biaxial mismatch stress to produce the dislocation HL.

During the numerical solution of Eqs. (1)-(5), criterion (14) is controlled at every time step. Initially, as the interface coordinate  $z$  is not too high, there are no V-shaped dislocation HLs ( $\rho = 0$ ) and Eq. (14) is not satisfied at any  $z_0$ . However, at some (critical) coordinate  $z_c$ , criterion (15) becomes valid at some value of  $z_0$  and  $\rho = 0$ . At this moment, the dislocations are generated with the apex located at  $z_0$ . Further,  $z_0$  does not change, while the dislocation density  $\rho$  increases, following the increasing coordinate  $z$  and continuously satisfying Eq. (14).

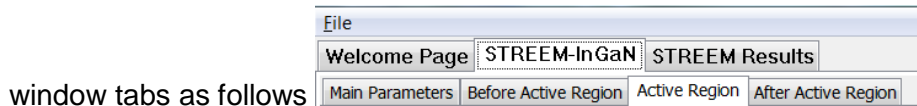
Eventually, the dislocation HL density can be found for every moment of time. Accounting for the fact that each dislocation HL has two wings and each of the wings represents a threading dislocation, the threading dislocation density is computed as

$$\rho_{TD} = 2 \cdot \rho + \rho_{TD}^0 \quad (16)$$

with  $\rho_{TD}^0$  being the user specified density of the threading dislocations in the layers underneath the active region.


## 4. Overview of the Graphical User Interface

Operation of the STREEM-InGaN software is implemented via Graphical User Interface (GUI) that is launched with the **STREEM-InGaN.exe** executable file. STREEM-InGaN GUI is aimed at working with the project files, specifying main computational parameters, defining process conditions, running computations, and viewing results. The STREEM-InGaN GUI contains the menu bar and a number of



window tabs as follows

- *File* menu contains several conventional options to work with the project files, as described in Section 5;
- **Welcome Page** tab allows the users to create a new project, open any existing project, and open a project from the list of the recently open projects. More details are given in Section 5;
- **STREEM-InGaN** tab contains a number of items, namely:
  - o **Main Parameters** tab to specify the governing parameters such as the surface segregation and stress relaxation models, units, and reactor transport model (see Section 6 for details);
  - o **Before Active Region**, **Active Region**, and **After Active Region** tabs to specify the recipe for the corresponding stages of the process, as discussed in Section 7;
- **STREEM Results** tab to visualize the modeling results, as discussed in Section 8.

To make the work with the STREEM-InGaN software more convenient, some features of the GUI can be adjusted by the user, depending on the particular settings of the computer (screen size and resolution, desktop font size, etc.). The font for the information specified in the recipe table can be changed by pressing  button and subsequent selection of the desired items in the standard Windows dialog shown in Fig. 4.1.

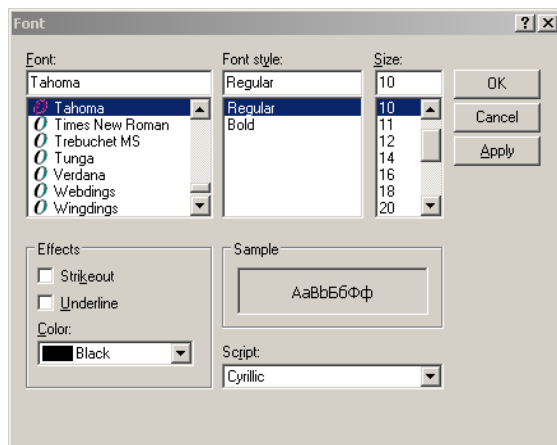


Fig. 4.1



Width of each particular column in the recipe table can be adjusted by dragging column border with the left mouse button. In case some information is beyond the screen, it can be approached with horizontal and vertical scrollbars. Another adjustable parameter is the number format during visualization of the results (see Section 8).

Once adjusted, the above GUI settings will be used in all subsequent launchings of the GUI, until any changes are introduced.



## 5. Working with the project files and Welcome Page

Information specified by the user within the STREEM-InGaN GUI is stored in a ***project\_name.stm*** file. Standard commands, including *New*, *Open*, *Save*, and *Save as...*, are available in the *File* menu bar of the GUI. Information stored in the ***project\_name.stm*** file can be easily loaded into the STREEM-InGaN GUI by pressing the file, if the files with the ***stm*** extension are associated with the GUI during software installation.

***Welcome Page*** is designed to provide quick access to the options of working with the project files. In particular, the user is allowed to open an existing project file with the *Open Project...* item on the left panel of the ***Welcome Page*** and start a new project with the *New Project...* item. In the latter case, the user is redirected to the ***Main Parameters*** tab of the GUI. On the ***Welcome Page*** it is also possible to load one of the recently opened projects from the respective list *Recently Opened Projects*; this list can be refreshed and non-existing files can be deleted from the list with the *Refresh* and *Delete Non Existing* items on the bottom panel of the ***Welcome Page***.

## 6. Specification of the main parameters

Before setting the growth recipe and making computations, the user is to specify some governing parameters on the **Main Parameters** tab of the GUI. These parameters include: selection of the surface segregation and stress relaxation models, choice of the units, and specification of the reactor transport model. Below, all these options are described in more detail.

### 6.1 Units

In this section, the user is to select, using the corresponding drop-down menu, the units for the temperature, pressure, growth rate, and group-III precursor flow rates. The selected units will be used throughout the project and can be changed on the **Main Parameters** tab only. Note that a change of a unit does not provide automatic re-calculation of the specified process parameters: e.g., if the pressure unit is changed from *Pa* to *mbar*, the operating pressure of 10,000 Pa will not automatically change to 100 mbar. The only exception is the growth rate for the layer outside the active region, as discussed in Section 7.3. The list of units allowed within the STREEM-InGaN is given in Table 6.1.

Process parameter	Units
Temperature	C, K
Pressure	Pa, Torr, mbar
Group-III flow rate	sccm, $\mu\text{mol}/\text{min}$
Growth rate	$\mu\text{m}/\text{h}$ , $\text{nm}/\text{sec}$

Table 6.1

### 6.2 Segregation Model

As described in Section 3, there are two options related to the surface segregation models included into the STREEM-InGaN software, which consist in the account of or ignoring surface site blocking with adsorbed indium. The model can be selected via the corresponding radio button in the *Segregation Model* section on the **Main Parameters** tab. The nominal composition profile, corresponding to the absence of any segregation effects, can also be simulated by selecting *No segregation* option.

### 6.3 Relaxation Model

Stress relaxation in the InGaN/GaN active region via formation of V-shaped dislocation half-loops can be considered by clicking the corresponding radio button in the *Relaxation Model* section on the **Main Parameters** tab. Otherwise, the active region is assumed to grow pseudomorphically on the underlying





layer (by default, GaN is assumed, but other layer can be specified by the user, as discussed in Section 7.3).

## 6.4 Reactor Transport Model

As discussed in Section 3, mass transport is treated within the boundary layer concept. There are three options to specify the characteristics of mass transport in the reactor, the active option is chosen by clicking the corresponding radio button in the *Reactor Transport Model* section:

- *Fixed diffusion layer thickness*: when selecting this option, the user is to specify directly the boundary layer thickness ( $\delta$ ) for each stage in the active region;
- *Calibration on thick GaN growth rate*: as a rule, the growth rate of a thick GaN layer is well known for some particular process parameters. If it is the case, this information can be used for calibration purpose. When *Calibration on thick GaN growth rate* item is selected, an additional section *Reactor Model* and a table for specifying GaN process conditions appear. Since the boundary layer thickness is computed differently for different reactor types, the reactor type should be selected via the proper radio button in the *Reactor Model* section. Here, we distinguish three types of the reactor: close coupled showerhead, planetary/horizontal, and rotating disk.

To set the reference GaN process conditions, the user is to fill the corresponding table. Here, the operating parameters are: temperature, pressure, carrier gas ( $N_2$  and  $H_2$ ) and ammonia flow rates, TMGa flow rate, and the reference growth rate. In case the rotating disk reactor, additional parameter, namely, the susceptor rotation rate (*rpm*) should be specified. Note that the TMGa flow rate (more general, group-III precursor flow rates for the active region) can be specified either directly or using bubbler characteristics, as described in Section 7.2.

The reference GaN process parameters are used to estimate the boundary layer thickness at the stage of thick GaN growth (reference thickness). The latter quantity is then scaled at each stage of the heterostructure growth in the active region with the account for the corresponding differences in process conditions.

- *Calibration on average InGaN growth rate and composition*: selection of this option assumes that the user will have to specify an average growth rate and composition for each layer in the active region, so that the boundary layer thickness will be adjusted to get these values in the modeling. Presently, the software utilizes the partial GaN growth rate ( $Vg^{GaN} \sim (1 - X_{In}) \cdot Vg^{InGaN}$ ) to estimate the boundary layer thickness, so that the resulting indium content may be somewhat different from the value specified by the user.



General view of the *Reactor Transport Model* section with specified process parameters for reference GaN is shown in Fig. 6.1.

Reactor transport model					Reactor model		
<input type="radio"/> Fixed diffusion layer thickness					<input checked="" type="radio"/> Close coupled showerhead		
<input checked="" type="radio"/> Calibration on thick GaN layer growth rate					<input type="radio"/> Horizontal/planetary reactor		
<input type="radio"/> Calibration on average InGaN growth rate and composition					<input type="radio"/> Rotating disk reactor		

Temperature C	Pressure Torr	N2 Flow R slm	H2 Flow Rate slm	NH3 Flow Rate slm	TMGa Flow Rate, $\mu\text{mol}/\text{min}$		Reference Growth Rate, $\mu\text{m}/\text{h}$
					Bub.	Given	
1000	75	0	5	3	<input type="checkbox"/>	130	3

**Fig. 6.1**



## 7. Specification of the recipe

Within the STREEM-InGaN software, the overall recipe is divided into three groups: stages (or layers) before the active region, active region, and stages after the active region. Parameters for these groups are specified on the respective tabs of the GUI.

To remind, an InGaN-based active region implies a sequence of quantum wells and barriers as well as other stages and special procedures in-between, like cap layer, pre-deposition, growth interruption, temperature ramping, etc. Modeling of the stress and composition profile is actually performed for the active region, so that it is important to specify correct process parameters for each stage of the active region. As a separate task, stress relaxation in a thick InGaN layer or technological superlattice can be considered by defining such layer as the active region (see Example 5).

The recipe is represented as a table; depending on the layer position, each stage is described by one (for the layers outside the active region) or two lines (for the layers in the active region) in the table. The number of columns (actually, of the parameters to be specified) is also different for the layers within or outside the active region, as discussed below.


For each stage of the process, the first column in the table is the *Repeat Count* flag (indicates that the stage belongs to a group), the second one is the *Stage Number*, and the third one is the stage *Name* (can be optionally set by the user). Meaning of other parameters in the table is explained below. General view of the recipe table for the layers before/after the active region and inside the active region is shown in Fig. 7.1. The active line of the table is marked with blue; the active cell is marked with yellow. In case of a wrong value (e.g., negative flow rate) is specified, this cell is colored with red.

Welcome Page   STREEM-InGaN   STREEM Results																				
Main Parameters   Before Active Region   Active Region   After Active Region																				
Repeat Count	Stage Number	Name	Thickness characteristics						Composition						Dislocation density 1/cm <sup>2</sup>		Stress state			
			Thickness,nm		Duration,sec		Growth rate,µm/h		AlN		GaN		InN		Inherited	Given	Used	a,Å	Used	Relax.Degree
			Calc	Given	Calc	Given	Calc	Given	Calc	Given	Calc	Given	Calc	Given						
1	n-GaN		450	810	2	0	1	0							1e10		3.189			
Main Parameters   Before Active Region   Active Region   After Active Region																				
Repeat Count	Stage Number	Name	Thickness characteristics						Composition						Dislocation density 1/cm <sup>2</sup>		Stress State			
			Thickness,nm		Duration,sec		Growth rate,µm/h		AlN		GaN		InN		Inherited	Given	Used	a,Å	Used	Relax.Degree
			Calc	Given	Calc	Given	Calc	Given	Calc	Given	Calc	Given	Calc	Given						
1	AlGaN		30	540	0.2	0.2	0.8	0										0.5		
2	p-GaN		100	200	1.8	0	1	0										1		
Main Parameters   Before Active Region   Active Region   After Active Region																				
Repeat Count	Stage Number	Name	Duration sec	Diffusion layer thickness,cm		Temperature C	Pressure Torr	N2 FlowRate slm	H2 FlowRate slm	NH3 FlowRate slm	TMIn FlowRate,µmol/min		TMGa FlowRate,µmol/min		TEGa FlowRate,µmol/min					
											Bub.	Given	Bub.	Given	Bub.	Given				
																	Init	Final	Init	Final
10	1	QW	210	1.1		780	350	4	0	-4		2.2		6		0				
						780	350	4	0	-4		2.2		6		0				
	2	barrier	230	1.1		Init	780	350	4	0	4		0		6		0			
						Final	780	350	4	0	4		0		6		0			

Fig. 7.1

## 7.1 Working with the process stages: general features

Regardless of the position of the particular layer in the heterostructure (within or outside the active region), there are several options that facilitate the procedure of recipe specification:

- There are some operations (grouping, deleting, and copying) that can be applied to several stages. They can be selected by clicking *Shift* + left mouse button.
- Grouping of the stages: recipe of a device heterostructure often includes a sequence of stages (QW+barrier, layers in a superlattice, etc.) that is repeated more than once. Such stages can be specified once and grouped with an assigned number of repetitions. To do this, the user is to select several stages and then press  button. Then a dialog window (Fig. 7.2) appears, where it is necessary to set the number of repetitions (**Repeat Count** field). Another opportunity is to select only one layer from the desired group and then define the stages to be grouped using *From* and *to* boxes. Grouping can be annulled by the same procedure with pressing **Clear** button at the end. Layers/stages merged into a group are indicated in the *Repeat Count* column in the recipe table.

Repeat Count	Stage Number	Name	Duration sec	Diffusion layer thickness,cm		Temperature C	Pressure Torr	N2 FlowRate slm	H2 FlowRate slm	NH3 FlowRate slm
20	1	QW	210	1.1	Init	780	350	4	0	4
					Final	780	350	4	0	4
	2	barrier	230	1.1	Init	780	350	4	0	4
					Final	780	350	4	0	4



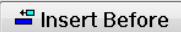


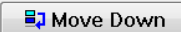
  

Repeat Count ☒



Stages  
From  to

Repeat Count

Fig. 7.2

- To append a stage, use  button, a blank line for a new process stage will be added at the end of the stage list.
- One or several stages can be deleted with  button.
- To insert an additional line in the recipe before or after the active (marked with blue) stage, use  and  buttons, correspondingly.
- Currently selected process stage can be moved up or down by one position with  and  buttons.



- One or several stages in the recipe can be copied. To do this, the user is to copy the selected stages using  button, than create a blank line in the proper position of the recipe using the *Append*, *Insert Before* or *Insert After* commands, and paste the selected stages using  button (the numbers of the lines to be pasted are shown above the button). Note that all the parameters of the selected stages will be copied, except for the stage names.

## 7.2 Stages in the active region

To describe a process stage in the active region, the user is to define the stage name (optional, but recommended item) and duration (*sec*). Next, it is necessary to specify the transport characteristics. Depending on the option selected in the *Reactor Transport Model* section on the **Main Parameters** tab, there are three possibilities:

- If the *Fixed diffusion layer thickness* option was chosen, the boundary layer thickness (*cm*) should be defined for each process stage in the corresponding field;
- If the *Calibration on thick GaN growth rate* option was chosen, there is no need to add any additional information;
- If the *Calibration on average InGaN growth rate and composition* option was chosen, the growth rate and composition should be assigned to each layer in the active region.

Next, the user is to specify the process parameters that include: temperature, pressure, flow rates of  $N_2$ ,  $H_2$ ,  $NH_3$ , and group-III precursors. Here, two lines (*Init* and *Final*) for these parameters are reserved, thus allowing linear variation of any operating parameter during the particular process stage. When the initial value is introduced, it is automatically copied as the final one, so, for the majority of cases, it is enough to insert the value only once.

It is recommended to avoid strictly stepwise changes in the process parameters between stages. For instance, if a barrier or cap layer (with zero TMIn flow rate) is grown directly after a QW (with some non-zero TMIn flow rate), it is preferable to introduce a short stage where the TMIn flow rate decreases from the value at the QW stage to zero, as represented by “TMIn-off” stage in Fig. 7.3. Adding this stage reflects the fact that the stepwise change of the gas composition at the reactor inlet proves somewhat smoothed when it reaches the growth surface due to the gas species interdiffusion. Simultaneously, this reduces the stiffness of the problem and improves stability of computations. Another typical case is temperature variation during the growth; here, the ramping time is normally known and the ramping can be specified as a recipe stage.

Repeat Count	Stage Number	Name	Duration sec		Temperature C	Pressure Torr	N2 FlowRate slm	H2 FlowRate slm	NH3 FlowRate slm	TMIn FlowRate, μmol/min		TMGa FlowRate, μmol/min	
										Bub.	Given	Bub.	Given
10	1	QW	210	Init	780	350	4	0	4	<input type="checkbox"/>	6	<input type="checkbox"/>	6
				Final	780	350	4	0	4	<input type="checkbox"/>	6	<input type="checkbox"/>	6
	2	TMIn-off	0.5	Init	780	350	4	0	4	<input checked="" type="checkbox"/>	6	<input checked="" type="checkbox"/>	6
				Final	780	350	4	0	4	<input checked="" type="checkbox"/>	0	<input checked="" type="checkbox"/>	6
	3	barrier	230	Init	780	350	4	0	4	<input type="checkbox"/>	0	<input type="checkbox"/>	6
				Final	780	350	4	0	4	<input type="checkbox"/>	0	<input type="checkbox"/>	6

Fig. 7.3

The MO flow rates can be specified either directly or using bubbler parameters. In the latter case, the bubbler configuration window appears, as shown in Fig. 7.4, after selecting *Bub.* checkbox. The first section of the dialog contains bubbler conditions: flow rate of the carrier gas passing through the bubbler  $F_{car}$ , bubbler pressure  $P$ , and temperature  $T$ .

The second section of the dialog controls the approximation used for the computation of the MO vapor pressure, which is required to compute the MO flow rate. The number of the available approximations depends on the MO species considered; however, there is also *User Defined* item that allows the user to specify custom approximation coefficients. The data for the approximation named *O.Kayser et al.* are taken from paper [7]. The data for the approximation named *Rohm and Haas* were borrowed in the past from the web. Regardless of the selected approximation, the MO vapor pressure  $P^o$  (in *Torr* or *mbar*, depending on the source of the approximation data) is computed as  $\log(P^o) = -A/T + B$ , where the approximation coefficients  $A$  and  $B$  are either built-in or defined by the user. Eventually, the MO flow rate is computed as  $F_{MO} = \frac{F_{car}}{P/P^o - 1}$  and displayed in the recipe table after pressing *OK* button.

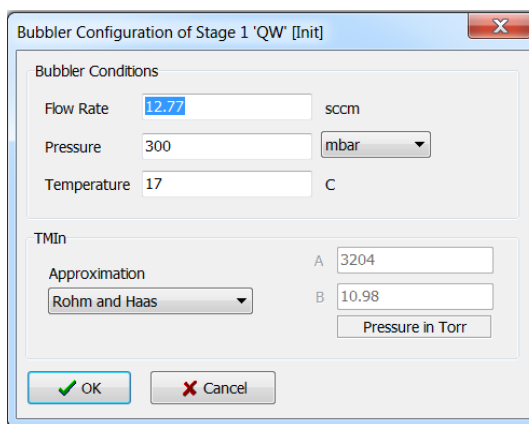
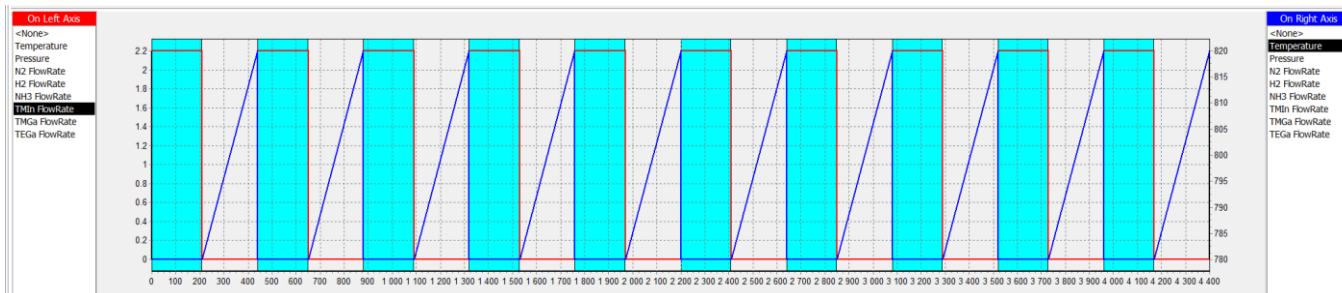


Fig. 7.4

To specify the MO flow rate directly instead of defining it from the bubbler parameters, just deselect the respective checkbox; to view and modify the bubbler parameters, use *Ctrl* + left mouse button.

For control, the process parameters, specified for the active region, are presented graphically at the bottom part of the **Active Region** tab. One of the process parameters can be selected for the left axis of

the plot (visualized in red), another one – for the right axis (visualized in blue). The currently active stage in the table is marked on the plot area as a single highlighted zone or several highlighted zones if the stage belongs to a group (see Fig. 7.5 as an example).



**Fig. 7.5**

This tool can be helpful for quick checking of the recipe to find possible inaccuracies in the specified process conditions.

Another parameter that may be required for modeling is the dislocation density. Depending on the *Relaxation Model* option and information provided for the layer before the active regions, there are several variants:

- If stress relaxation via V-shaped dislocation half-loops is considered, there is no need to specify the dislocation density in the active region, since this parameter will be computed;
- If stress relaxation is not considered, the dislocation density ( $1/cm^2$ ) for each layer in the active region should be either defined directly or inherited from the previous layer. This is made by selecting the corresponding checkbox in the recipe table, as demonstrated in Fig. 7.6. If there are no layers before the active region, the dislocation density should be specified at least for the first stage in the active region. Actually, when no stress relaxation is assumed, the specified dislocation density does not affect the computed composition profile and serves as supplementary information that can be subsequently used for device modeling with the SiLENSe software.

Dislocation density, $1/cm^2$	
Inherited	Given
<input type="checkbox"/>	1e9
<input type="checkbox"/>	1e9
<input checked="" type="checkbox"/>	
<input checked="" type="checkbox"/>	
<input checked="" type="checkbox"/>	
<input checked="" type="checkbox"/>	

**Fig. 7.6**

### 7.3 Layers before and after the active region

Since the modeling of composition profile is performed for the active region only, the process parameters for the layers outside it are not required. Instead, four sections should be filled in to specify the layers before and after the active region. The parameters described below are assumed to remain unchanged within each layer outside the active region.

- **Thickness characteristics:** here, three parameters are available for editing: *Thickness*, *Duration*, and *Growth rate*; the user is to define an arbitrary pair of them, while the third one is automatically restored. The latter parameter is indicated by *Calc* radio button in the respective column, and this field is marked with green, as displayed in Fig. 7.7. The two other parameters are inserted into the *Given* field of the respective columns.

Repeat Count	Stage Number	Name	Thickness characteristics					
			Thickness, nm		Duration, sec		Growth rate, $\mu\text{m/h}$	
			Calc	Given	Calc	Given	Calc	Given
	1	n-GaN	<input type="radio"/>	450	<input checked="" type="radio"/>	810	<input type="radio"/>	2

Fig. 7.7

If the user changes the growth rate unit in the *Units* section on the **Main Parameters** tab, the table for the layers before and after the active region is automatically renewed in the following way: (i) if the growth rate is the given parameter, its value remains the same, but the calculated value changes due to the different unit of the growth rate; (ii) if the growth rate is the calculated parameter, its value changes, but the two other parameters remain the same. The user is informed about the changes with the warning window shown in Fig. 7.8;

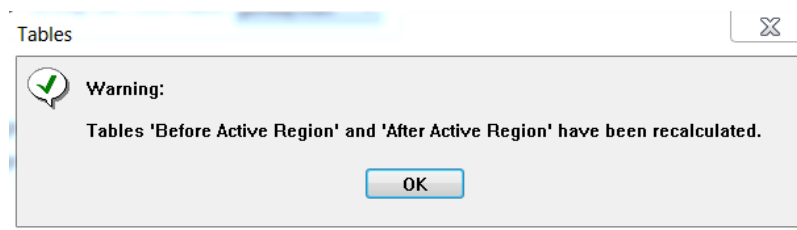


Fig. 7.8

- **Composition:** similarly to the thickness characteristics, the layer composition is defined by the content of any two constituents of an AlInGaN layer, for the remaining constituent, the content is recalculated from the condition of the unity total content;
- **Dislocation density:** can be specified directly or inherited from the previous layer; for the first layer before the active region, the dislocation density should be defined explicitly;
- **Stress state:** one of the purposes of including layers before the active region is to specify the lattice constant of the layer underlying the active region to subsequently use this value for lattice mismatch



and strain computations. The stress state can be defined either via the  $a$  lattice constant ( $\text{\AA}$ ) or via the relaxation degree. The relaxation degree serves to find the effective lattice constant of the layer

and is determined as  $r = \frac{a_{current}^{eff} - a_{previous}^{eff}}{a_{current}^{bulk} - a_{previous}^{eff}}$ , where  $a_{current}^{eff}$  and  $a_{previous}^{eff}$  are the effective lattice

constants of the current and previous layers, respectively, and  $a_{current}^{bulk}$  is the lattice constant of the bulk (fully relaxed) material. Thus, for the first layer in the heterostructure, only the lattice parameter can be specified.

The lattice parameters of the binary materials, used in the STREEM-InGaN software, are summarized in Table 7.1. Vegard's law is applied to compute the lattice parameters for ternary and quaternary layers.

Material	$a$ , $\text{\AA}$
AlN	3.112
GaN	3.189
InN	3.54

Table 7.1

Example of layers before the active region is shown in Fig. 7.9: thick p-GaN is followed by a relaxed  $\text{In}_{0.08}\text{Ga}_{0.92}\text{N}$  underlayer whose lattice parameter will be used in the active region (more details on the role of underlayers can be found in the description of Example 4, Section 9).



Name	Thickness characteristics						Composition						Dislocation density $1/\text{cm}^2$		Stress State			
	Thickness, nm		Duration, sec		Growth rate, $\mu\text{m/h}$		AlN		GaN		InN		Inherited	Given	Used	$a$ , $\text{\AA}$	Used	Relax. Degree
	Calc	Given	Calc	Given	Calc	Given	Calc	Given	Calc	Given	Calc	Given						
n-GaN	○	450	○	810	*	2	*	0	○	1	○	0	□	1e10	*	3.189	○	
underlayer	○	180	*	3240	○	0.2	○	0	*	0.92	○	0.08	*		○		*	1

Fig. 7.9

Lattice constant or relaxation degree for the layers after the active region can be specified in the same way. If the relaxation degree is defined for a layer after the active region, the effective lattice constant at the end of the active region growth is used as that of the previous layer. The effective lattice constant is one of the variables available during visualization of the results, as discussed in the next Section.

Note that the layers before and after the active region can be omitted in the recipe. In this case, the active region is assumed to grow on a thick relaxed GaN.

## 8. Running the computations and viewing the results

To run the computations, press  button that is available in the bottom section of **Before Active Region**, **Active Region**, and **After Active Region** tabs. The solver progress is depicted with  window. In case of any solver errors, a window appears as shown in Fig. 8.1. The most probable solver error is too small time step. If so, a possible solution may be modifying the recipe to avoid stepwise changes in the process parameters as discussed in Section 7. In case of normal termination of the solver, the user is informed about this and redirected to the **STREEM Results** tab (see Fig. 8.1). Modeling results are stored in the **project\_name.dat** file and loaded automatically for visualization on the **STREEM Results** tab.

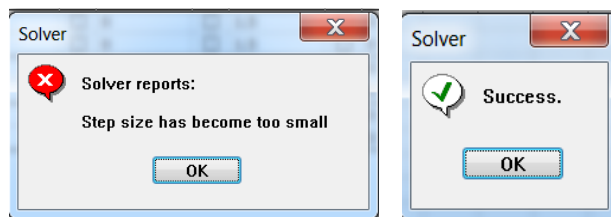
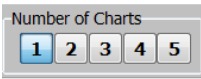
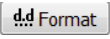



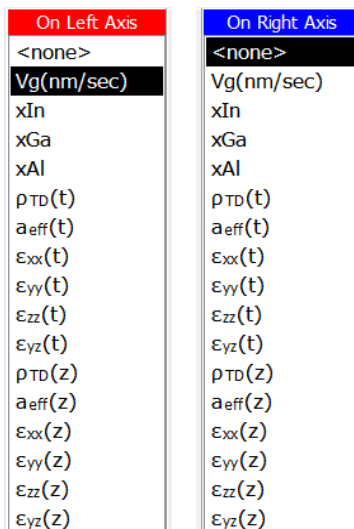
Fig. 8.1

Results of the computations are visualized in form of one-dimensional plots. As the x-axis variable, the user can select either *time (sec)* or interface coordinate *z (nm)* from *Argument* drop-down menu. Each plot may have one or two y-axes, and the dependent variable for the left and right axis is selected from the respective sections on the **STREEM Results** tab as shown in Fig 8.2. Red and blues curves represent the left-axis variable and the right-axis one, correspondingly. Up to five charts can be

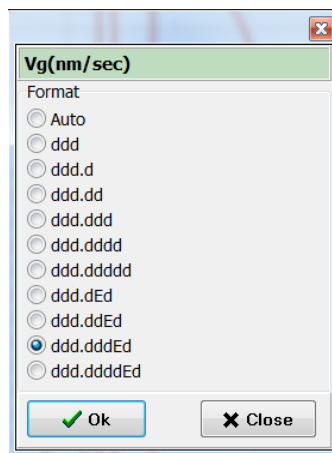
displayed simultaneously; the number of charts is selected with  section.

To adjust the label format for any particular variable, use one of  buttons and then select the format from the available list as demonstrated in Fig. 8.3.


The numerical technique implemented in the STREEM-InGaN software utilizes automatic adaptation of time steps, and it is possible to follow time step variation during the computations. For this purpose, points can be added on the plots, using  section on the **STREEM Results** tab. The first button is to display lines only; the other buttons correspond to the points of different size on the plots. Close location of points over some part of the curve means that the solver had to reduce the time step considerably due to steep gradients of some variables.

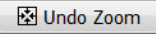


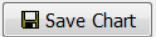
**Fig. 8.2**


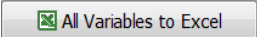


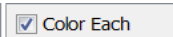
**Fig. 8.3**

At any cursor position on the chart field, the user can see the values of the x-axis and y-axis variables above the plot area: `time(sec)=676.41417 Vg(nm/sec)=0.047 xIn=0.04541`. For better representation of the probe position, press  button.

To zoom in a part of the graph, select the area of interest on the chart by pressing the left mouse button and dragging the mouse to a diagonal corner of the desired area, the overall direction should be left-to-right; the selected area will be represented by a black rectangle. To zoom out, drag the mouse in the opposite direction or press  button.

Each chart field can be exported into an image file by pressing  button and further selection of the file format and name in a standard dialog window.

For each chart, the currently displayed variables can be exported to an Excel sheet with  button. All computational results can be exported by pressing  button.

Since the actual composition profile and thicknesses of individual layers in the heterostructure are different from the nominal values due to the surface segregation effects and stress relaxation, it is important to distinguish particular stages of the recipe in the plots. For this purpose, all layers from the recipe are listed in the *Layers* section on the **STREEM Results** tab. By activating  checkbox, it is possible to associate the layers, having the same name, with particular color. One or several layers can be selected by toggling corresponding checkboxes. Fig. 8.4 shows an example where all quantum wells in the structure are selected. If so, the selected layers are marked on the chart area with highlighted zones which are either cyan (if the *Color Each* checkbox is inactive) or have the color specific for this group of layers. Example of visualization with layers highlighted in both ways is given in Fig. 8.5. Note that an arbitrary number of layers with arbitrary names (ultimately, all layers) can

be highlighted at a time. The buttons in  section are to select all layers, deselect all layers, and invert selection.

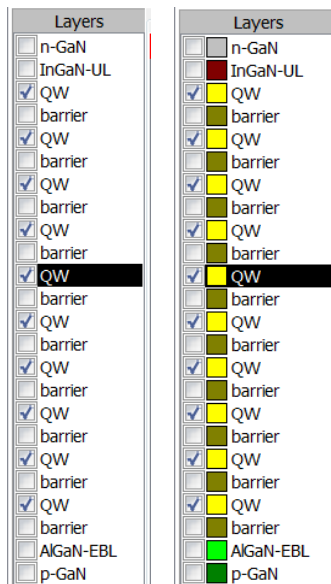
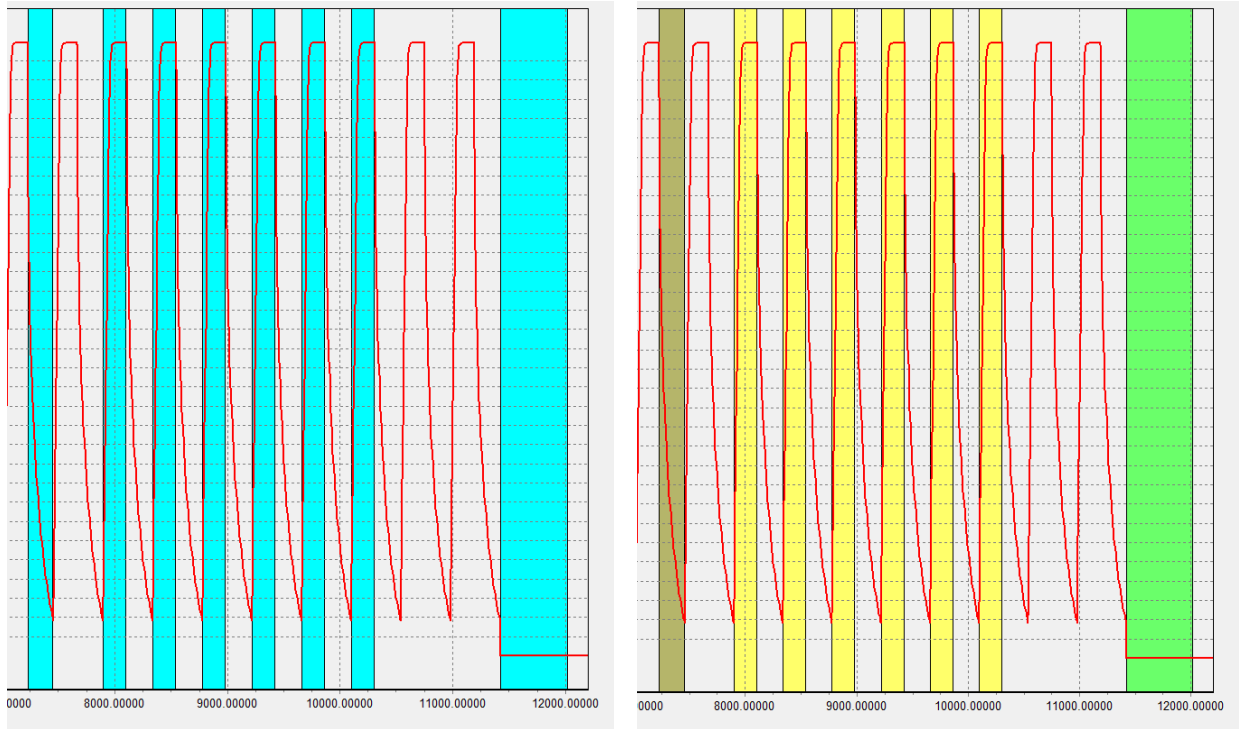


Fig. 8.4



**Fig. 8.5**

The list of dependent variables with the respective units is given in Table 8.1.

Variable	Meaning	Unit
$Vg$	growth rate	$nm/sec$
$xIn, xGa, xAl$	layer composition	-
$\rho_{TD}(t)$	threading dislocation density vs time	$cm^{-2}$
$a_{eff}(t)$	effective lattice constant vs time	$\text{Å}$
$\varepsilon_{xx}(t), \varepsilon_{yy}(t), \varepsilon_{zz}(t), \varepsilon_{yz}(t)$	strain tensor components vs time	-
$\rho_{TD}(z)$	threading dislocation density vs thickness	$cm^{-2}$
$a_{eff}(z)$	effective lattice constant vs thickness	$\text{Å}$
$\varepsilon_{xx}(z), \varepsilon_{yy}(z), \varepsilon_{zz}(z), \varepsilon_{yz}(z)$	strain tensor components vs thickness	-

**Table 8.1**

The effective lattice constant is computed as  $a_{eff} = a^{bulk} \cdot (\varepsilon_e + \varepsilon_p + 1)$ , where  $a^{bulk}$  is the lattice constant of bulk InGaAlN with the composition  $xIn$ ,  $xGa$ , and  $xAl$ ,  $\varepsilon_e(z) = \frac{a_s - a^{bulk}}{a^{bulk}}$  is the elastic deformation ( $a_s$  is the latticed parameter of the underlying layer), and  $\varepsilon_p$  is the plastic deformation that depends on the

dislocation density. Until stress relaxation occurs, the effective lattice constant remains unchanged and equal to  $a_s$ , in the opposite ultimate case of a fully relaxed layer,  $a_{eff} = a^{bulk}$ .

For [0001] growth direction, the strain tensor components are:  $\varepsilon_{xx} = \varepsilon$ ,  $\varepsilon_{yy} = \varepsilon$ ,  $\varepsilon_{zz} = -2 \frac{C_{13}}{C_{33}} \varepsilon$ ,  $\varepsilon_{yz} = 0$ ,

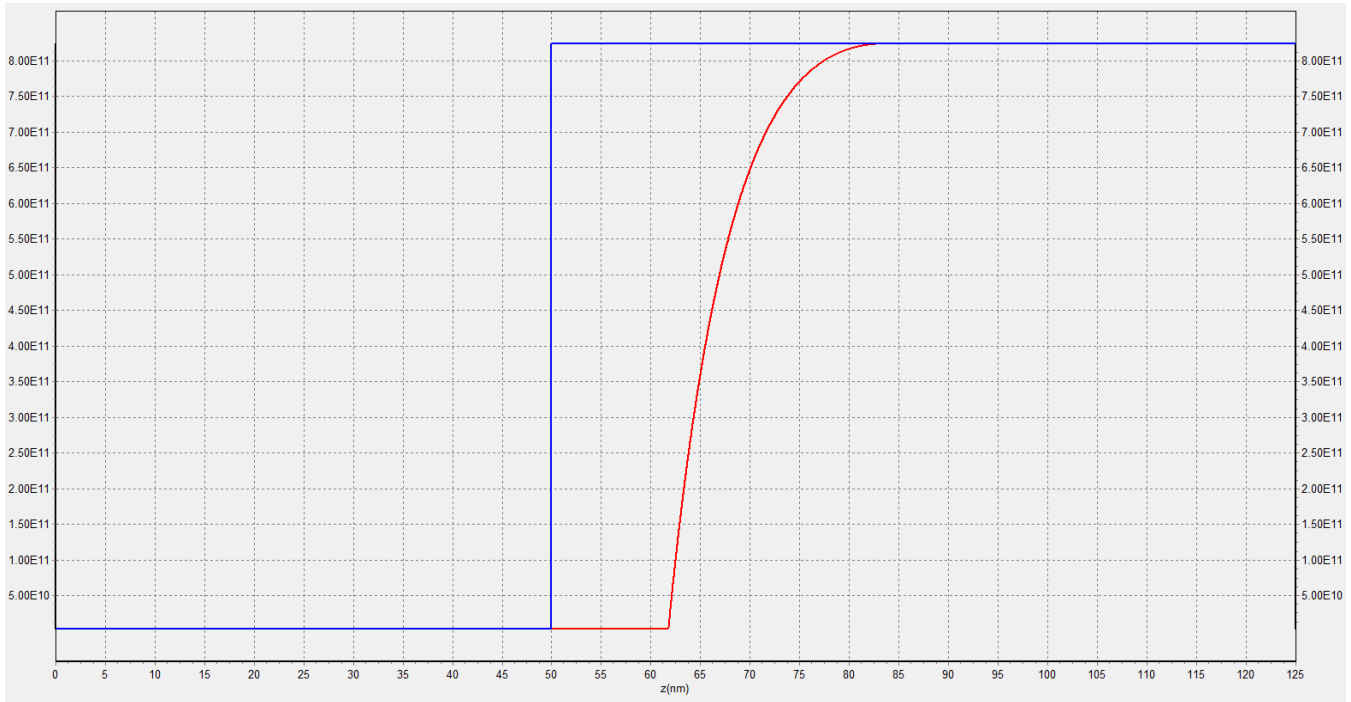
where  $\varepsilon = \frac{a^{eff} - a^{bulk}}{a^{bulk}}$  is the total deformation and  $C_{ij}$  are the elastic constants.

It is important to note that the dislocation density, effective lattice constant, and strain tensor components are included into the list of variables as functions of both time and thickness. All characteristics of the heterostructure are primarily found in STREEM-InGaN as functions of time. For each moment of time  $t_i$ ,  $z(t_i)$  is the local coordinate of the upper interface of the heterostructure at that time. This means that the distributions of any variable versus time and coordinate represent the evolution of in-situ values of this variable and are formed as  $t_i, z(t_i), value(t_i)$ .

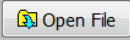
However, from practical point of view, it is also important to know resulting distributions of the above characteristics across the structure **after** the structure growth. For instance, measurements of the lattice parameter with XRD or observations of dislocations with TEM are normally performed after the growth. Modeling of device operation also requires spatial rather than temporal variations of the heterostructure characteristics. Thus, the final distributions of some variables versus thickness are additionally introduced in the STREEM-InGaN.

The main reason for differences between the temporal (in-situ) and spatial (resulting) variations comes from the behavior of the dislocations. As discussed in Section 3 and references cited there, new dislocation half-loops are produced during the growth, but, being generated, they instantly penetrate down, as a rule, to the InGaN/GaN interface (the apex coordinate  $z_0$  is found during the computations). As a result, the threading dislocation density across the thickness varies in a stepwise mode, from the value before the active region (specified by the user) to the value predicted by the modeling. Comparison of the  $\rho_{TD}(t)$  (red curve) and  $\rho_{TD}(z)$  (blue curve) distributions for a thick relaxed InGaN layer is shown in Fig. 8.6: while  $\rho_{TD}(t)$  monotonically increases,  $\rho_{TD}(z)$  increases sharply and remains constant, corresponding to the maximum value of  $\rho_{TD}(t)$ .

Following the dislocation density, the related characteristics, such as the effective lattice constant and strain, also have different distributions versus time and thickness when stress relaxation occurs. In contrast, the distribution of the layer composition versus time and thickness are similar, as the chemical composition may change only locally and these changes do not affect the layers underneath.



**Fig. 8.6**

By default, the results of the current computation are visualized. Results, corresponding to other projects, can be loaded using  button on the **STREEM Results** tab.

STREEM-InGaN output file **project\_name.dat** can be directly imported into the SiLENSe software. The composition, strain, and dislocation density distributions computed within the STREEM-InGaN will be applied for the modeling of LED characteristics with the SiLENSe.

## 9. Examples

The examples below illustrate the application of the STREEM-InGaN software to the analysis of the influence of indium surface segregation, process parameters, stress relaxation, and thick InGaN undelayers on the resulting characteristics of InGaN-based heterostructures. The examples are stored in the *Examples* folder. To see the main parameters and recipes, load the **project\_name.stm** file from the respective sub-folder (*Example1÷Example5*), to see the computational results, run the computation or load the result file **project\_name.dat**.

### 9.1 Example 1: segregation effects in SQW structure

The example considers a rather simple case when the growth of an InGaN QW is followed by the growth of a GaN barrier layer. Here, the project files **SQW.stm** and **SQW\_no\_segregation.stm** correspond to the computation with the segregation effect and with *No Segregation* option in the *Segregation Model* section on the **Main Parameters** tab. Parameters of GaN growth in a vertical high-speed rotating disk reactor are specified as reference ones (see *Reactor Transport Model* section). Figs. 9.1-9.2 allow one to understand the effect of the indium surface segregation on the indium composition profile (nominal QW region is marked with cyan): while the computation with no segregation demonstrates a rectangular QW (Fig. 9.1), the computation considering this effect (Fig. 9.2) predicts an indium tail in the barrier after TMIn supply is terminated.

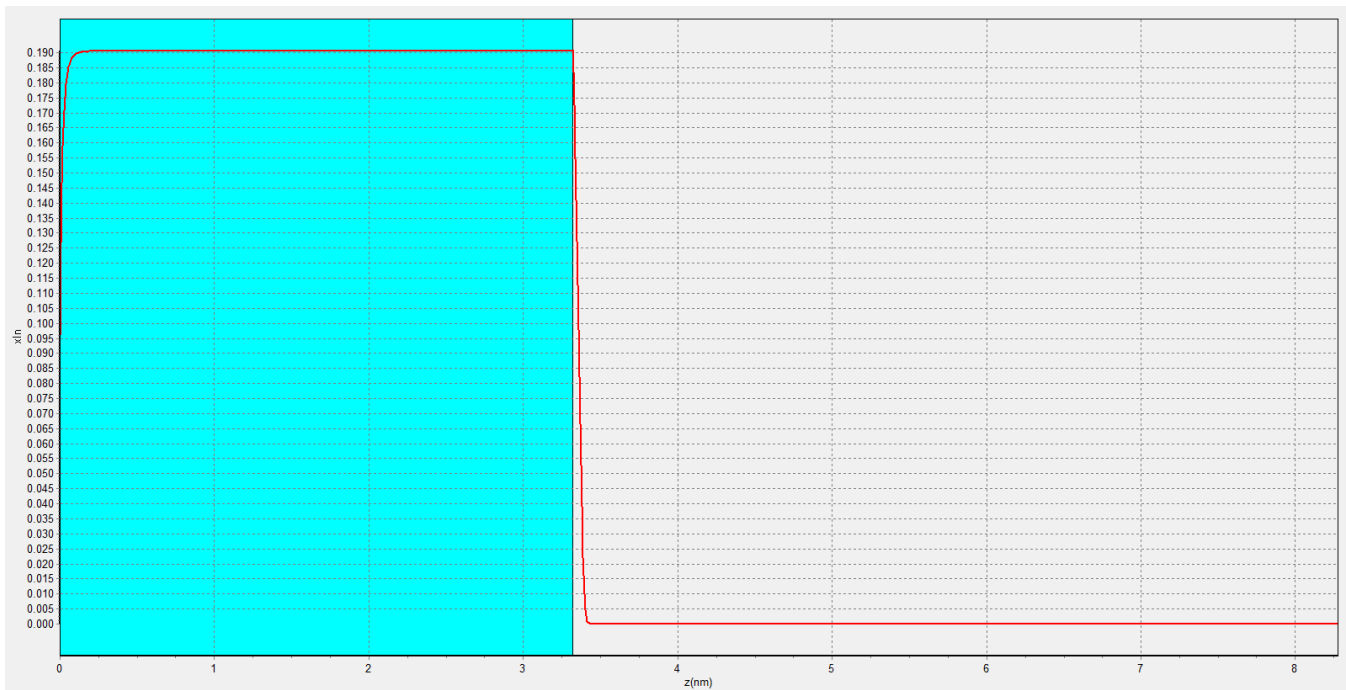
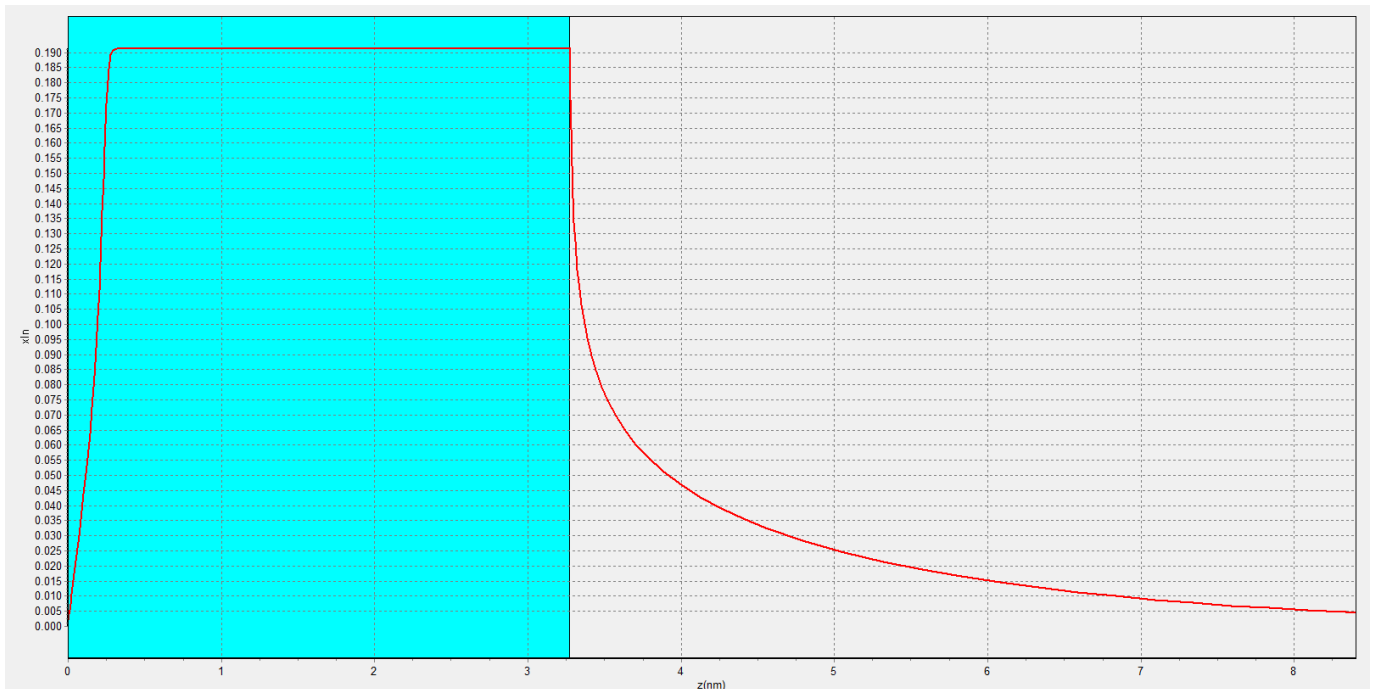


Fig. 9.1





**Fig. 9.2**

## ***9.2 Example 2: effect of process parameters on indium composition profile in MQW structure with moderate indium content***

The example illustrates the effect of process parameters on the indium incorporation and composition profile across a MQW InGaN/GaN structure. Growth of five InGaN QWs separated by GaN barriers is considered. As the transport model, the diffusion layer thickness is directly specified for the stages in the active region. The reference process parameters (corresponding to the 2<sup>nd</sup> QW) are similar to those presented in Ref [3]:

- InGaN QW growth temperature – 750 °C;
- GaN cap growth temperature – 750 °C;
- GaN barrier growth temperature – 920 °C;
- Total pressure – 200 Torr;
- TMIn flow rate – 0.4 sccm;
- TEGa flow rate – 0.8 sccm;
- NH<sub>3</sub> flow rate – 30 slm;
- N<sub>2</sub> flow rate – 60 slm;
- QW growth duration – 90 sec;
- Barrier growth duration – 180 sec.



The stages of QW and barrier growth (*QW\_Growth* and *Barrier\_Growth*, respectively) are separated by the intermediate stages of GaN cap growth (*Cap\_Growth*, 180 sec), temperature ramping (*T\_RampUp* and *T\_RampDown*, 90 sec), and temperature stabilization (*T\_Stab*, 90 sec). In addition, for better computation stability, each QW growth stage is separated from the previous and next stages by very short stages of TMIn switching-on and -off (*In\_RampUp* and *In\_RampDown*, 1 sec). A part of the recipe table, corresponding to reference period of the MQW structure, is given in Fig. 9.3.

Stage Number	Name	Duration sec	Diffusion layer thickness,cm		Temperature C		Pressure Torr	N2 FlowRate slm	H2 FlowRate slm	NH3 FlowRate slm	TMIn FlowRate,sscm		TMGa FlowRate,sscm		TEGa FlowRate,sscm	
					Init	Final					Bub.	Given	Bub.	Given	Bub.	Given
9	In_RampUp	1	0.325		Init	750	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
					Final	200	60	0	30	<input type="checkbox"/>	0.4	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8	
10	QW_Growth	90	0.325		Init	750	200	60	0	30	<input type="checkbox"/>	0.4	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
					Final	200	60	0	30	<input type="checkbox"/>	0.4	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8	
11	In_RampDown	1	0.325		Init	750	200	60	0	30	<input type="checkbox"/>	0.4	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
					Final	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8	
12	Cap_Growth	180	0.325		Init	750	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
					Final	750	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
13	T_RampUp	90	0.325		Init	750	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
					Final	920	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
14	Barrier_Growth	180	0.325		Init	920	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
					Final	920	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
15	T_RampDown	90	0.325		Init	920	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
					Final	750	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
16	T_Stab	90	0.325		Init	750	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8
					Final	750	200	60	0	30	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0.8

Fig. 9.3

Within the example, the following parameters are varied as compared to the reference case:

- Temperature of InGaN QW growth – 700 °C (1<sup>st</sup> QW);
- Temperature of GaN cap layer – 750-920 °C (3<sup>rd</sup> QW);
- TMIn flow rate – 0.7 sccm (4<sup>th</sup> QW);
- TEGa flow rate – 0.4 sccm (5<sup>th</sup> QW).

The corresponding In content profile computed with the software is presented in Fig. 9.4. Here, the nominal InGaN QW growth stages, GaN cap growth stages, and temperature ramping stages are shown in brown, pink, and blue colors, respectively; the other stages are not indicated with any color.

It is seen from the figure that the computed In content profile essentially differs from the nominal one. In particular, it exhibits abrupt but still somewhat sloped fore fronts and considerably sloped rear fronts, so that rather long indium tails prove incorporated into the GaN barriers. Depending on the process parameters, the computed profile demonstrates the following effects:

- Decrease of InGaN QW growth temperature from 750 °C to 700 °C results in increase of the In content from ~16% to ~26% (2<sup>nd</sup> reference QW vs. 1<sup>st</sup> QW);
- Shortening of the low-temperature (750 °C) part of GaN cap growth stage from 180 sec to 30 sec provides shortening of the In “tail” in the barrier from ~4 nm to ~2 nm (3<sup>rd</sup> QW vs. reference 2<sup>nd</sup> QW);
- Rise of TMIn flow rate from 0.4 sccm to 0.7 sccm results in increase of the In content from ~16%

to ~20% (4<sup>th</sup> QW vs. reference 2<sup>nd</sup> QW);

- Decrease of TEGa flow rate from 0.8 sccm to 0.4 sccm leads to almost proportional shortening of the InGaN QW thickness, from ~2 nm to ~1 nm, but weakly affects the In incorporation (5<sup>th</sup> QW vs. reference 2<sup>nd</sup> QW)

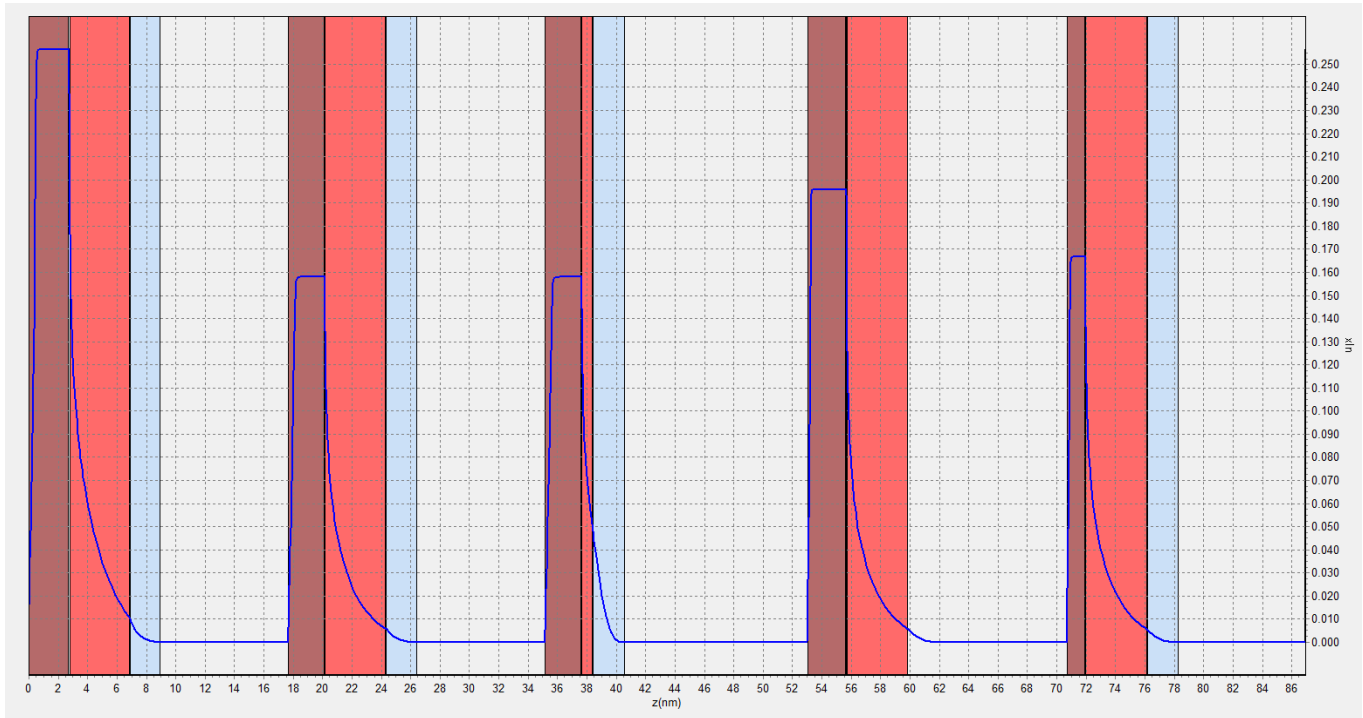


Fig. 9.4

### 9.3 Example 3: MQW structure with stress relaxation

The so called ‘green gap’ in the efficiency of III-nitride light-emitting diodes and laser diodes is frequently attributed, at least in part, to intensive generation of extended defects during stress relaxation in active (0001)InGaN quantum wells grown on thick GaN buffer/contact layers. In the following example case, a MQW structure is analyzed with respect to stress relaxation in the active region and respective changes in the indium composition profile. The structure is similar to that reported in Refs. [8-9], and the active region consists of five periods, each of them includes the following stages: quantum well, cap layer, growth interruption accompanied by temperature ramping from  $T_{QW}$  to  $T_{barrier}$ , barrier, and the second temperature ramping (from  $T_{barrier}$  back to  $T_{QW}$ ). The process conditions are applicable to AIX2000HT planetary reactor, and calibration on thick GaN growth is used. The process conditions for one period of the recipe are given in Fig. 9.5.

Repeat Count	Stage Number	Name	Duration sec		Temperature C	Pressure mbar	N2 FlowRate slm	H2 FlowRate slm	NH3 FlowRate slm	TMIn FlowRate, sscm		TMGa FlowRate, sscm		TEGa FlowRate, sscm	
										Bub.	Given	Bub.	Given	Bub.	Given
5	1	MO-on	0.5	Init	685	200	10	0	23	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0
				Final	685	200	10	0	23	<input checked="" type="checkbox"/>	0.64048	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	0.41343
	2	QW	60	Init	685	200	10	0	23	<input checked="" type="checkbox"/>	0.64048	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	0.41343
				Final	685	200	10	0	23	<input checked="" type="checkbox"/>	0.64048	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	0.41343
	3	TMIn-off	0.5	Init	685	200	10	0	23	<input checked="" type="checkbox"/>	0.64048	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	0.41343
				Final	685	200	10	0	23	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	1.0459	<input type="checkbox"/>	0
	4	cap	14	Init	685	200	10	0	23	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	1.0459	<input type="checkbox"/>	0
				Final	685	200	10	0	23	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	1.0459	<input type="checkbox"/>	0
	5	MO-off	0.5	Init	685	200	10	0	23	<input checked="" type="checkbox"/>	0	<input checked="" type="checkbox"/>	1.0459	<input type="checkbox"/>	0
				Final	685	200	10	0	23	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	1.0459	<input type="checkbox"/>	0
	6	ramping-up	40	Init	685	200	10	0	23	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0
				Final	750	200	10	0	23	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0
	7	TMGa-on	0.5	Init	750	200	10	0	23	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0
				Final	750	200	10	0	23	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	1.0459	<input type="checkbox"/>	0
	8	barrier	200	Init	750	200	10	0	23	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	1.0459	<input type="checkbox"/>	0
				Final	750	200	10	0	23	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	1.0459	<input type="checkbox"/>	0
	9	TMGa-off	0.5	Init	750	200	10	0	23	<input type="checkbox"/>	0	<input checked="" type="checkbox"/>	1.0459	<input type="checkbox"/>	0
				Final	750	200	10	0	23	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0
	10	ramping-down	40	Init	750	200	10	0	23	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0
				Final	685	200	10	0	23	<input type="checkbox"/>	0	<input type="checkbox"/>	0	<input type="checkbox"/>	0

Fig. 9.5

The example includes two project files: **MQW\_no\_relaxation.stm** one is for *No Relaxation* option, **MQW\_with\_relaxation.stm** one is for *V-dislocations* option in the *Relaxation Model* section. To model the indium surface segregation, the “standard” approach with no site blocking effect is applied, which seems preferable for the structures with enhanced indium content.

Fig. 9.6 presents the indium content distribution in the active regions. When the stress relaxation is ignored (*No Relaxation* option), the indium content remains the same, regardless the number of QWs. The account of stress relaxation via formation of V-shaped dislocation HLs (*V-dislocations* option) modifies the composition profile: the stress relaxation occurs after the second QW, and the indium incorporation increases subsequently in the 3<sup>rd</sup>-5<sup>th</sup> quantum wells. This results correlates with experimental data which demonstrate an increase in the emission wavelength and indium content as the number of QWs is raised from one to five.

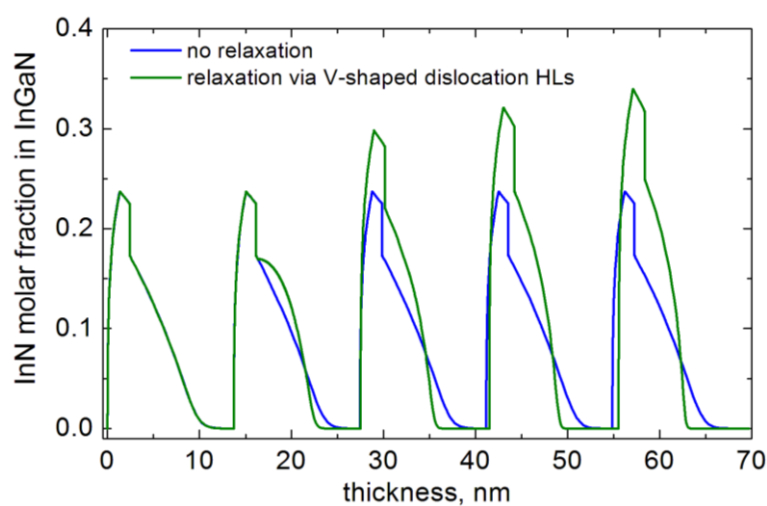


Fig. 9.6

## 9.4 Example 4: effect of InGaN underlayers

The use of InGaN underlayers (UL) and InGaN/GaN superlattices is now actively exploited during the growth of III-Nitride LED structures for strain balancing in the active regions of heterostructures. Addition an InGaN layer under the MQW stack was shown to improve the efficiency of the MQWs. The example below demonstrates how the effect of ULs can be interpreted within the STREEM-InGaN software.

This example considers the growth of so-called strain balanced multiple quantum well (SBMQW) structures consisting of ten  $\text{In}_x\text{Ga}_{1-x}\text{N}(4\text{nm})/\text{GaN}(4\text{nm})$  periods grown on 180 nm thick  $\text{In}_y\text{Ga}_{1-y}\text{N}$  UL for  $x>y$  [10]. Following [10], the  $\text{In}_y\text{Ga}_{1-y}\text{N}$  ULs are assumed to be fully relaxed. The SBMQW active region implies that  $x=2y$ , if the wells and barriers have the same thickness. Indium composition in the quantum wells was controlled by changing the growth temperature. Since process parameters are not exactly known, a GaN growth regime in the 3x2" close-coupled showerhead reactor is chosen as the reference.

$\text{In}_x\text{Ga}_{1-x}\text{N}/\text{GaN}$  MQW structures with  $x=16\%$  and  $x=22\%$  grown on  $\text{In}_y\text{Ga}_{1-y}\text{N}$  underlayers with  $y=8\%$  and  $y=11\%$ , respectively, are compared to the conventional structures without UL. The project files **InGaN-UL-x0.08.stm** and **InGaN-UL-x0.11.stm** are for the ULs with  $y=8\%$  and  $y=11\%$ , files **InGaN-x0.08.stm** and **InGaN-x0.11.stm** are for the same structures without ULs. The structure begins with an n-GaN layer, followed by fully relaxed InGaN for the structures with the UL, and is supplemented with partly relaxed AlGaIn blocking layer and p-GaN one after the active region. Process recipe before the active region and for the active region is presented in Fig. 9.7 for  $y=8\%$ . Throughout the computations, the *V-dislocations* option was selected in the *Relaxation Model* section to account for stress relaxation.

Repeat Count	Stage Number	Name	Thickness characteristics						Composition						Dislocation density 1/cm <sup>2</sup>		Stress state			
			Thickness,nm		Duration,sec		Growth rate, $\mu\text{m}/\text{h}$		AlN		GaN		InN		Inherited	Given	Used	a, $\text{\AA}$	Used	Relax.Deg
			Calc	Given	Calc	Given	Calc	Given	Calc	Given	Calc	Given	Calc	Given						
	1	n-GaN	○	300	○	540	●	2	○	0	○	1	○	0	□	1e+9	●	3.189	○	
	2	InGaN-UL	○	180	●	6480	○	0.1	●	0	○	0.92	○	0.08	■		○		●	1

Repeat Count	Stage Number	Name	Duration sec		Temperature C	Pressure Torr	N2 FlowRate slm	H2 FlowRate slm	NH3 FlowRate slm	TMIn FlowRate, $\mu\text{mol}/\text{min}$		TMGa FlowRate, $\mu\text{mol}/\text{min}$		TEGa FlowRate, $\mu\text{mol}/\text{min}$	
										Bub.	Given	Bub.	Given	Bub.	Given
10	1	QW	210		780	350	4	0	4	■	2,2008	■	6,0037	□	0
					780	350	4	0	4	■	2,2008	■	6,0037	□	0
10	2	barrier	230		780	350	4	0	4	□	0	■	6,0037	□	0
					780	350	4	0	4	□	0	■	6,0037	□	0

Fig. 9.7

The use of the underlayer allows the growth of a ten period MQW structure with the same indium composition profile across the quantum wells and no stress relaxation. In contrast, stress relaxation after several starting QWs, formation of new dislocations in the active region, and changes in the In content distribution in the subsequent QWs are predicted for the structures without UL, as demonstrated in Fig. 9.8 for the active region of the structures with different indium content. These findings are in agreement with the data given in [10]. In particular, the authors report that the ten period structures can

be grown without relaxation-related effects, using the SBMQW growth technique.

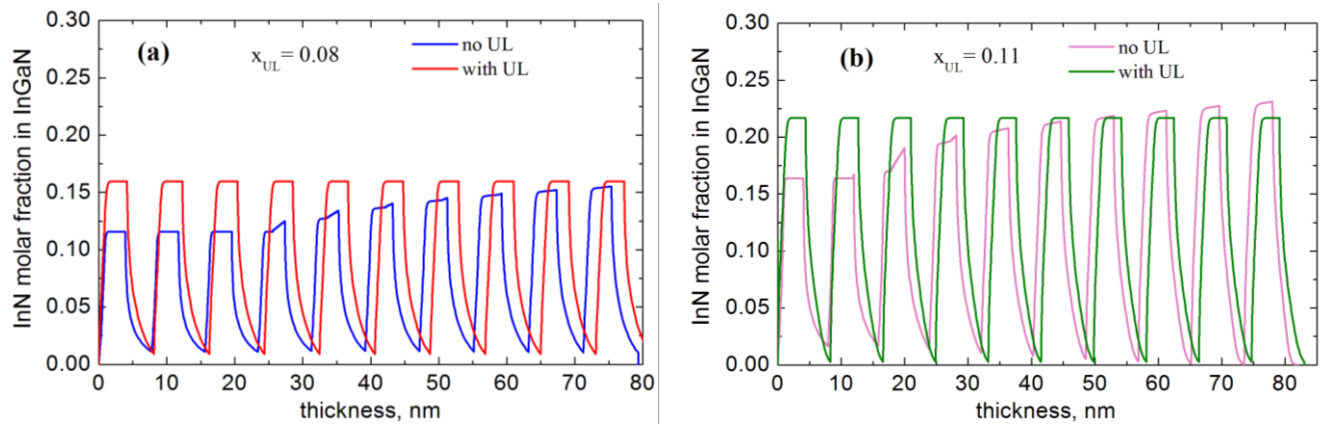


Fig. 9.8

### 9.5 Example 5: relaxation of thick InGaN layer

The previous example case assumes a 180 nm thick fully relaxed InGaN UL before the active region. The authors of [10] claim that the ULs with the thickness of 80 nm did not relax, while 160 nm and 180 nm thick ULs were relaxed. The results presented in Fig. 9.9 agree with these observations: InGaN layer with the initial indium content of about 6 % starts to relax at the thickness of about 30 nm (in agreement with the data on the critical thickness vs composition from [5]) and reaches the constant composition of about 8 % when its thickness exceeds 90 nm.

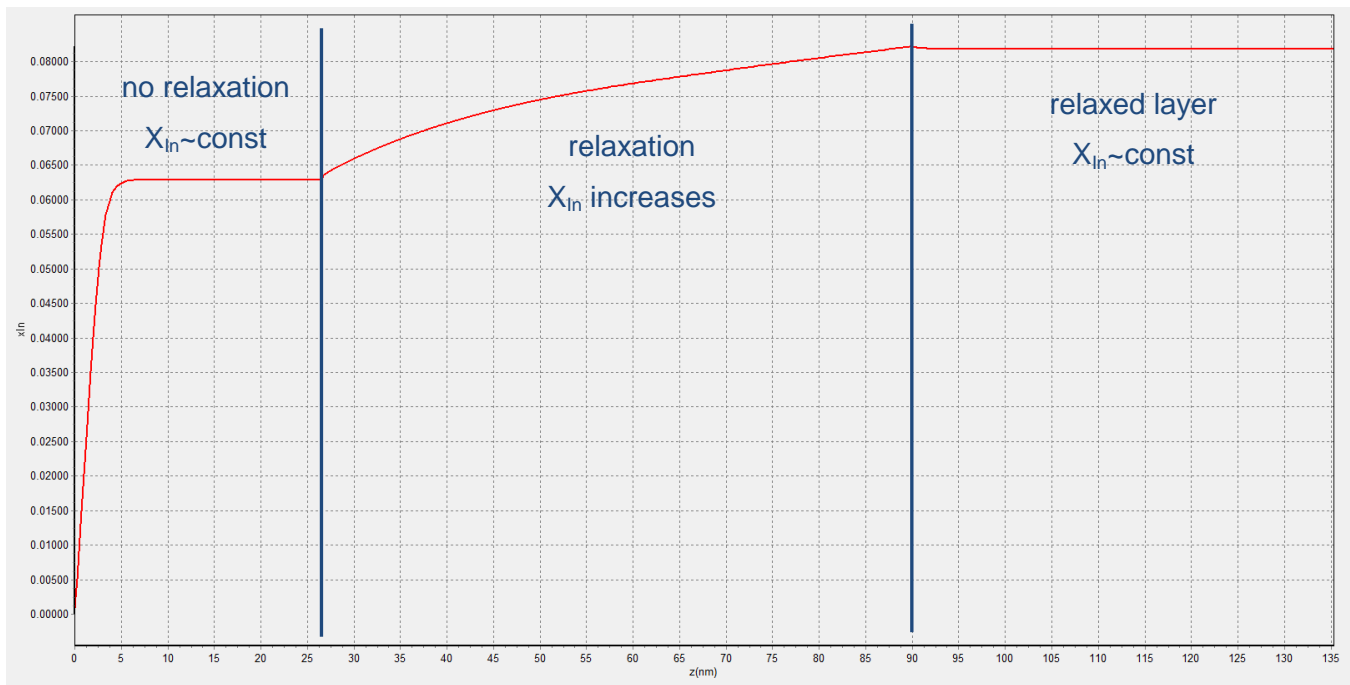


Fig. 9.9



## 10. Limitations

1. The current version of the software supports InGaN-based active regions only with arbitrary indium content. AllInGaN layers can be specified outside the active region only in case such layers are present in the heterostructure under consideration.
2. Presently, annihilation of threading dislocations is not considered in the model. We suppose to introduce this option in the next release of the STREAM-InGaN software.

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